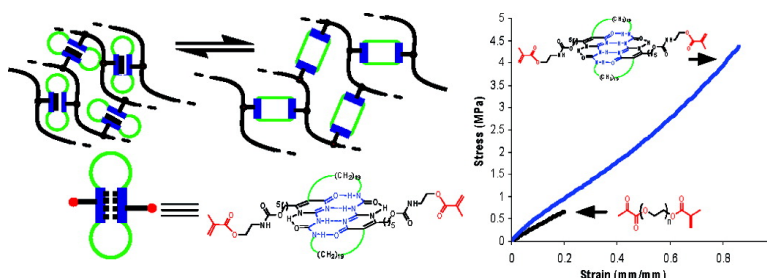


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## Biomimetic Design of Reversibly Unfolding Cross-Linker to Enhance Mechanical Properties of 3D Network Polymers

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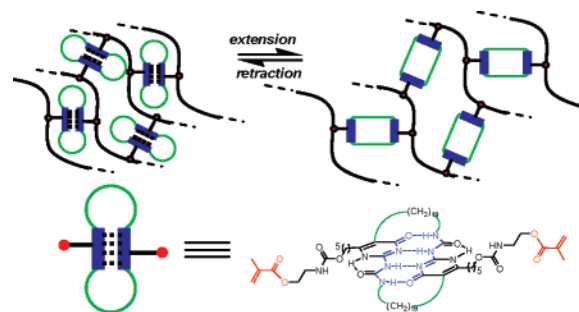
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Three-dimensional (3D) network polymers are an important family of materials. For many applications of 3D networks, it is important to combine high modulus, high tensile strength, and high extensibility.<sup>1</sup> Rigid network materials tend to fail after only a short extension. While flexible elastomers are more extensible, they usually have low moduli and exhibit shallow stress response. Although many engineering approaches and chemical modifications<sup>2</sup> have been developed to improve mechanical properties of network polymers, it remains a challenge to design ideal networks that have a combination of desired properties. Among chemical methods, interesting work has been reported on using interchain hydrogen bonding to improve polymer physical properties.<sup>3</sup> Given the importance of cross-linker structure on mechanical properties of network polymers, it is surprising that there is very limited investigation on designing molecularly engineered cross-linkers to enhance network properties. Herein we introduce a novel biomimetic design of a reversibly unfolding modular cross-linker that can increase elastomer stiffness without sacrificing extensibility, leading to a dramatic tensile strength enhancement.

Our biomimetic concept is based on the modular design observed in many biopolymers, such as the skeletal muscle protein *titin* and connective proteins in both soft and hard tissues, that have a remarkable combination of strength and elasticity.<sup>4</sup> Single-molecule nanomechanical studies have revealed that their combined mechanical properties originate from their unique modular structure, which sequentially unfolds upon deformation providing the molecular mechanism to sustain high force (strength) and to yield high elongation (elasticity).<sup>5</sup> Inspired by nature, our group has been mimicking this modular domain strategy in the pursuit of synthetic polymers with advanced mechanical properties. A number of biomimetic modules have been successfully designed and incorporated into linear polymers.<sup>6,7</sup> Single-molecule and bulk properties validated our biomimetic concept of using modular structures to enhance polymer properties. This communication describes the first example of introducing a reversibly unfolding modular cross-linker into 3D networks to enhance their mechanical properties.

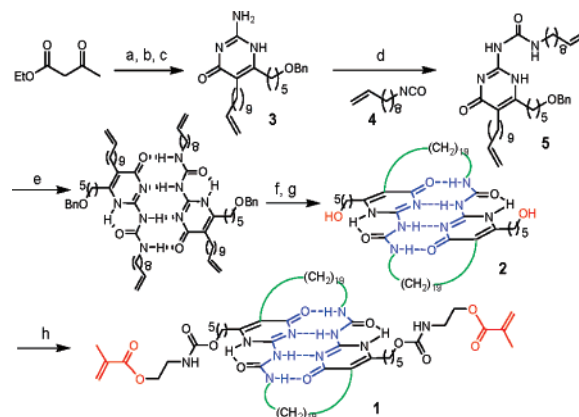
Our concept is illustrated in Figure 1. A stress applied from any direction to a 3D network will be ultimately transferred across the individual network junctions, where biomimetic modules can be reversibly unfolded. Since it requires significant forces to unfold the modules held by strong multiple hydrogen bonds, further extension can be gained without sacrificing the modulus. In addition, we propose that the enhanced energy dissipation capability by unfolding the biomimetic modules should lead to significant increases in both extensibility and tensile strength. In this study, we use a simple poly(*n*-butyl acrylate) network to demonstrate this concept.

The biomimetic cross-linker is based on the quadruple hydrogen bonding 4-ureido-2-pyrimidone (UPy) motif originally developed by the Meijer and co-workers.<sup>8</sup> The synthesis of the module is shown in Scheme 1 (also see synthetic details in Supporting Information). Double alkylation of ethyl acetoacetate followed by guanidine condensation afforded the alkenyl-pyrimidone interme-



**Figure 1.** Concept of biomimetic design of biomimetic modular cross-linker for enhancing network mechanical properties.

### Scheme 1. Synthesis of the Biomimetic UPy Modular Cross-Linker<sup>a</sup>



<sup>a</sup> Conditions: (a) NaH, *n*-BuLi, THF, Br(CH<sub>2</sub>)<sub>4</sub>OBn, 52%; (b) Br(CH<sub>2</sub>)<sub>9</sub>CH=CH<sub>2</sub>, K<sub>2</sub>CO<sub>3</sub>, DMF, rt, 12 h, 47%; (c) guanidine carbonate, EtOH, reflux, 12 h, 72%; (d) **4**, Py, reflux, 6 h, 83%; (e) DCM, reflux, 1–2 h, 65%; (f) (Cy<sub>3</sub>P)<sub>2</sub>RuCl<sub>2</sub>=CHPh, DCM, reflux, 1–2 h, 65%; (g) Pd(OH), 1 atm H<sub>2</sub>, THF, 68%; (h) CHCl<sub>3</sub>, 2-isocyanatoethyl methacrylate, DBTDL.

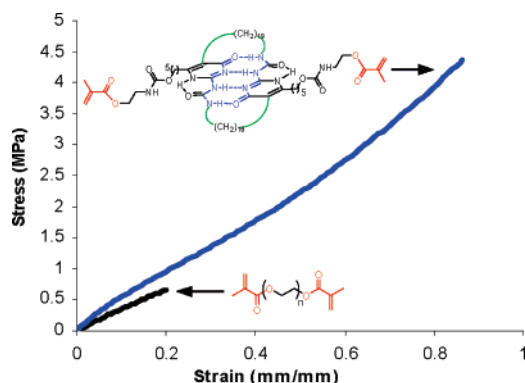
diate **3**. The isocyanate **4** was coupled to the pyrimidone **3** to yield **5**. Upon dimerization in DCM, ring-closing metathesis (RCM) effectively cyclizes the two UPy units.<sup>9</sup> Despite the possibility of other isomer formation, only the desired product was obtained (see Supporting Information). A one-pot reduction and deprotection through hydrogenation gave the diol module **2**. Finally, capping **2** with 2-isocyanatoethyl methacrylate at both ends provided the UPy cross-linker, **1**, which was characterized by <sup>1</sup>H and <sup>13</sup>C NMR, FTIR, and MALDI-TOF MS (see Supporting Information).

In this model study, we chose poly(*n*-butyl acrylate) as the backbone because this polymer exhibits elastomeric properties upon cross-linking (*T*<sub>g</sub> = −54 °C)<sup>10</sup> and can be easily synthesized by free radical polymerization. A series of transparent rubbery films were prepared by copolymerization of *n*-butyl acrylate with the cross-linker **1** using AIBN as initiator. A poly(ethylene glycol) (PEG) dimethacrylate (*M*<sub>n</sub> = 750) was used as the control cross-linker. This cross-linker is very soluble and flexible, and when fully stretched, it has approximately the same length as the fully unfolded

**Table 1.** Stress–Strain Data for Both Sample and Control Systems

sample	Young's Modulus (E) (MPa)	tensile strength (MPa)	elongation at failure (mm/mm)
PEG (2%)	1.24 ( $\pm 0.07$ )	0.52 ( $\pm 0.04$ )	0.59 ( $\pm 0.06$ )
PEG (4%)	1.67 ( $\pm 0.12$ )	0.57 ( $\pm 0.06$ )	0.45 ( $\pm 0.05$ )
PEG (6%)	3.82 ( $\pm 0.13$ )	0.63 ( $\pm 0.17$ )	0.19 ( $\pm 0.05$ )
UPy (2%)	1.67 ( $\pm 0.09$ )	0.95 ( $\pm 0.12$ )	0.75 ( $\pm 0.07$ )
UPy (4%)	3.89 ( $\pm 0.02$ )	2.47 ( $\pm 0.38$ )	0.69 ( $\pm 0.05$ )
UPy (6%)	5.57 ( $\pm 0.08$ )	4.5 <sup>a</sup>	0.8 <sup>a</sup>

<sup>a</sup> The data for specimens that did not break within the 10 N load limit (strain rate: 100 mm/min, at room temperature).



**Figure 2.** Stress–strain curves for 6% cross-linked poly(*n*-butyl acrylate) rubber for the sample (blue) and control specimen (strain rate: 100 mm/min, at room temperature).

UPy cross-linker **1**. The thermosets were characterized by FTIR, DSC, and static stress–strain and dynamic mechanical analysis (DMA). The  $T_g$  values measured by DMA for controls and real samples are all below room temperature (see Figure S8 in Supporting Information).

Table 1 summarizes the mechanical parameters obtained from the stress–strain studies, and Figure 2 compares the stress–strain curves for the sample and control specimen having 6 mol % of cross-linkers. Compared to the control, the key mechanical properties are significantly improved by introducing our biomimetic cross-linker. While the improvement is modest for the 2% samples, in the 4% UPy case, both modulus and tensile strength increase substantially without sacrificing the maximum elongation. At 6% incorporation, the UPy sample maintains a continuously improved modulus and a greater than 700% increase in tensile strength while maintaining a similar level of elongation. The rubbery plateau moduli obtained from DMA measurements (see Figure S6 in Supporting Information) agree well with the static tensile data.

The comparison between the UPy samples and the PEG controls validates our biomimetic concept. For the PEG controls, the increase of modulus at higher cross-linker levels trades off with the maximum elongation. This is typical for regular thermoset elastomers: higher cross-linking density results in more rigid and less elastic rubbers.<sup>2,11</sup> In contrast, in our UPy samples, we observed a consistent increase in modulus and tensile strength without sacrificing the extensibility. We attribute this to the increased energy dissipation ability of our modular cross-linker.<sup>12</sup> While the modulus is increased at higher cross-linking density, the reversibly unfolding modules can act as energy dissipating units to prevent fracture formation. To extend the control polymer (an entropic elastomer) by 100%, the free energy change ( $\Delta G$ ) is roughly  $-T\Delta S$ , which is estimated to be  $\sim 0.6$  kcal/mol.<sup>13</sup> In contrast, it takes  $\sim 11$  kcal/mol to fully unfold the UPy folded module (based on  $K_d \sim 10^{-8}$  M in toluene).<sup>8</sup> The enhancement of energy dissipation by the modular

cross-linker is supported by the much larger loss moduli measured by DMA for the real samples as compared to the controls (see Figure S7). Whereas potential nano- and microphases may also contribute to the bulk mechanical properties, on the basis of the optical transparency of the samples and our small-angle X-ray scattering (SAXS) and scanning electron microscopy (SEM) data (Figures S9 and S10), it is unlikely that nano- or microphases exist in our system.

In summary, we describe here the biomimetic design of a reversibly unfolding modular cross-linker to enhance mechanical properties of 3D networks. A UPy-based cyclic modular cross-linker was synthesized via multistep organic synthesis. Stress–strain measurements show that the poly(*n*-butyl acrylate) samples containing the modular cross-linker exhibit significantly enhanced mechanical properties over the control samples. Most interestingly, with increasing cross-linker density, both modulus and tensile strength are significantly improved without sacrificing the extensibility. This introduces a novel biomimetic concept to enhance rubber properties through design of molecularly engineered cross-linkers. Further studies are currently ongoing to extend this concept to other network systems and to probe the mechanisms for the property enhancement.

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**Supporting Information Available:** Synthesis and characterization of cross-linker and polymers, MALDI-TOF MS, MTS stress–strain experiments. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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